

III-NITRIDE COMPOUND SEMICONDUCTOR LIGHT EMITTING DEVICE

[Technical Field]

The present invention relates to a III-nitride compound semiconductor
5 light emitting device, and more particularly, to a III-nitride compound
semiconductor light emitting device with improved external quantum efficiency,
which has roughness in a nitride semiconductor layer.

As used herein, the term III-nitride compound semiconductor light
emitting device refers to a light emitting device, such as a light emitting diode
10 comprising a compound semiconductor layer made of $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$
($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$), and does not exclude the inclusion of either
materials made of other group elements, such as SiC, SiN, SiCN, and CN, or
a semiconductor layer made of such materials.

[Background Art]

15 FIG. 1 shows a III-nitride compound semiconductor light emitting
device according to the prior art. As shown in FIG. 1, the light emitting device
comprises: the substrate 100; the buffer layer 200 epitaxially grown on the
substrate 100; the n-type nitride semiconductor layer 300 epitaxially grown on
the buffer layer 200, the active layer 400 epitaxially grown on the n-type nitride
20 layer 300; the p-type nitride semiconductor layer 500 epitaxially grown on the
active layer 400; the p-side electrode 600 formed on the p-type nitride
semiconductor layer 500; the p-side bonding pad 700 formed on the p-side

electrode 600; and the n-side electrode 800 formed on the n-type nitride semiconductor layer 301 exposed by mesa-etching of at least the p-type nitride semiconductor layer 500 and the active layer 400.

The substrate 100 can use a GaN-based substrate as a homogeneous substrate, and a sapphire substrate, a silicon carbide substrate or a silicon substrate as a heterogeneous substrate, but can use any other substrates on which nitride semiconductor layers can be grown.

The nitride semiconductor layers epitaxially grown on the substrate 100 are usually grown by means of MOCVD (Metal Organic Chemical Vapor Deposition) method.

The buffer layer 200 serves to reduce differences in lattice constant and the coefficient of thermal expansion between the heterogeneous substrate 100 and the nitride semiconductor. U.S. Patent No.5,122,845 discloses a technology in which an AlN buffer layer having a thickness of 100 Å to 500 Å is grown on a sapphire substrate at a temperature ranging from 380°C to 800°C. U.S. Patent No.5,290,393 discloses a technology in which an $\text{Al}(x)\text{Ga}(1-x)\text{N}$ ($0 \leq x < 1$) buffer layer having a thickness of 10 Å to 5000 Å is grown on a sapphire substrate at a temperature ranging from 200°C to 900°C. Korean Patent No.10-0448352 discloses a technology in which a SiC buffer layer is grown at a temperature ranging from 600°C to 990°C, and an $\text{In}(x)\text{Ga}(1-x)\text{N}$ ($0 < x \leq 1$) layer is grown on the SiC buffer layer.

In the n-type nitride semiconductor layer 300, at least a region (n-type

contact layer) in which the n-side electrode 800 is formed is doped with an impurity. The n-type contact layer is preferably made of GaN and is doped with Si. U.S. Patent No.5,733,796 discloses a technology in which an n-type contact layer is doped with a desired doping concentration by controlling a
5 mixing ratio of Si and other source materials.

The active layer 400 is a layer for emitting a photon (light) by recombination of electrons and holes, and is mainly made of $\text{In}(x)\text{Ga}(1-x)\text{N}$ ($0 < x \leq 1$). The active layer 400 is composed of a single quantum well or multi quantum wells. WO02/021121 discloses a technology in which only some of a
10 plurality of quantum wells and barrier layers are doped.

The p-type nitride semiconductor layer 500 is doped with an impurity such as Mg, and has a p-type conductivity through an activation process. U.S. Patent No.5,247,533 discloses a technology in which a p-type nitride semiconductor layer is activated by means of irradiation of electron beam. U.S.
15 Patent No.5,306,662 discloses a technology in which a p-type nitride semiconductor layer is activated through annealing at a temperature of 400°C or more. Korean Patent No.10-043346 discloses a technology in which NH_3 and a hydrazine-based source material are used together as a nitrogen precursor for growing a p-type nitride semiconductor layer, so that the p-type
20 nitride semiconductor layer has a p-type conductivity without an activation process.

The p-side electrode 600 serves to allow the current to be supplied to

the entire p-type nitride semiconductor layer 500. U.S. Patent No.5,563,422 discloses a technology of a light-transmitting electrode, which is formed almost on the entire p-type nitride semiconductor layer, in ohmic contact with the p-type nitride semiconductor layer, and made of Ni and Au. U.S. Patent
5 No.6,515,306 discloses a technology of a light-transmitting electrode made of ITO(Indium Tin Oxide), which is formed on the n-type superlattice layer formed on the p-type nitride semiconductor layer.

Meanwhile, the p-side electrode 600 can be formed to have such a thick thickness that the p-side electrode 600 does not transmit light, i.e., the
10 p-side electrode 600 reflects light toward the substrate. A light emitting device using this p-side electrode 600 is called a flip chip. U.S. Patent No.6,194,743 discloses a technology of an electrode structure including an Ag layer of 20nm or more in thickness, a diffusion barrier layer covering the Ag layer, and a bonding layer made of Au and Al, which covers the diffusion barrier layer.

15 P-side bonding pad 700 and n-side electrode 800 are for providing current into the device and for wire-bonding out of the device. U.S. Patent No.5,563,422 discloses a technology of an n-side electrode made of Ti and Al. U.S. Patent No.5,652,434 discloses a technology of p-side bonding pad directly contacted with p-type nitride semiconductor layer by partially removing
20 the light-transmitting electrode.

One of unavoidable basic problems in the semiconductor light emitting device is that a phenomenon will occur where a significant portion of light generated in the semiconductor light emitting device is entrapped within the

semiconductor light emitting device due to a difference in dielectric constant between the semiconductor light emitting device and the surrounding air. The material of a III-nitride semiconductor light emitting device also has a refractive index of about 2.5 which is 1.5 greater than a refractive index of 1 for the surrounding air. Due to this difference in refractive index, a significant portion of light emitted from the active layer of the III-nitride semiconductor light emitting device is entrapped and disappears as heat in the light emitting device.

About 80% of light emitted from the active layer, as theoretically calculated, is entrapped and disappears in the light emitting device. To improve this light entrapment phenomenon is very important in that it can increase the external quantum efficiency of the light emitting device to maximize the output of the light emitting device.

In order to improve the external quantum efficiency, the chip configuration of the light emitting device is mechanically processed in the case of Cree Co. Also in some articles, the surface is made rough by a chemical etching or dry etching technique to improve the external quantum efficiency. Recently, the surface is also made rough by changing growth conditions, such as pressure, temperature and gas flow, in the growth of the p-type layer, while causing deterioration in the film quality. The above-mentioned mechanical processing is easy on a substrate, such as SiC, but almost impossible on a sapphire substrate with high strength. Also, the use of the chemical or dry etching technique has a limitation in area which can be made rough. Another

problem is that the process becomes complicated since an etching process is further added to the existing LED (light emitting diode) process. Also, there are problems in the reproduction and uniformity of the etching process. Rather than methods requiring this additional process, it is preferable in view of
5 a subsequent process to make the surface rough by changing growth conditions as described above. However, in the case of the method of making the surface rough by changing the growth conditions, the deterioration in the thin-film quality is unavoidable. Also, to achieve this improvement, a significantly thick layer must be formed. Accordingly, the growth of a layer
10 with bad quality in more than certain thickness can increase the external quantum efficiency of the light emitting device, but cause a fatal problem in the reliability of the light emitting device.

[Disclosure]

[Technical Problem]

15 It is an object of the present invention to provide a III-nitride compound semiconductor light emitting device whose external quantum efficiency can be increased without causing damages to the electrical properties of the light emitting device.

[Technical Solution]

20 To achieve this object, the present invention provides a III-nitride compound semiconductor light emitting device in which a first layer composed of a carbon-containing compound layer, such as an n-type or p-type silicon

carbide (SiC), silicon carbon nitride (SiCN) or carbon nitride layer (CN) layer, is formed on the p-type III-nitride semiconductor layer of the existing III-nitride semiconductor light emitting device, and a second layer composed of a III-nitride semiconductor layer with a given thickness is formed on the first
5 layer.

[Advantageous Effects]

According to the present invention, the external quantum efficiency of the light emitting device can be maximized by forming an effective roughness on an active layer by the peculiar method of the present invention. This is
10 because the light emitting device according to the present invention has a structure where light emitted from the active layer can be refracted on the roughness surface while effectively getting out of the light emitting device.

[Description of Drawings]

FIG. 1 is a view showing a III-nitride compound semiconductor light
15 emitting device according to the prior art;

FIG. 2 is a representative view showing one example of a light emitting device according to the present invention;

FIGs. 3 to 5 shows cross-sectional views of the first layer and the second layer formed in accordance with the present invention;

20 FIG. 6 shows scanning electron microscope (SEM) images of the surface of the device formed in accordance with the present invention;

FIGs. 7 and 8 show surface images of the device formed in accordance

with the present invention, measured by AFM equipment;

FIG. 9 is a graph showing an increase in light output with an increase in the thickness of the first layer of the inventive light emitting device; and,

FIG. 10 is photographs showing the surface roughness as a function of the thickness of the first layer of the present invention.

[Mode for Invention]

FIG. 2 is a representative view showing one example of a light emitting device according to the present invention. As shown in FIG. 2, the first layer 20 composed of a silicon carbide (SiC), silicon carbon nitride (SiCN) or carbon nitride (CN) layer is typically grown on the existing p-type GaN 14, on which the III-nitride semiconductor layer 21 with roughness (protrusions and/or depressions).

A method of forming the first layer composed of the silicon carbide (SiC), silicon carbon nitride (SiCN) or carbon nitride (CN) layer on the p-type $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer of the existing III-nitride semiconductor light emitting device will now be described in detail.

1) Formation of Si_aC_b layer

To form the SiC layer in the present invention, DTBSi, a kind of metal organic source, was used as a silicon source, and CBr_4 , a kind of metal organic source, was used as a carbon source. The use of these sources provides an advantage in that all the two sources are easily thermally decomposed at low temperature so that they can form the SiC layer at a lower temperature than

that in the use of a hydride source, such as CH_4 or SiH_4 . However, in the present invention, sources used in forming the SiC layer are not limited to DTBSi and CBr_4 . Important particulars in forming the SiC layer on the $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer are the growth
5 temperature and time of the SiC layer since a nitrogen source is not used during the growth of the SiC layer. If the growth temperature is excessively high or the growth time lengthens, a phenomenon will occur where N leave from the $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer so as to cause the aggregation of metals, such as Al, In and Ga. When this metal
10 aggregation phenomenon occurs, the growth of a high-quality thin film will become impossible and the characteristics (e.g., leakage current and reliability) of the resulting light emitting device will be adversely affected.

The SiC layer according to the present invention can be formed on the $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer in various forms,
15 such as amorphous, polycrystalline and monocrystalline forms, depending on conditions. Such crystalline forms can vary depending on the growth temperature of the SiC layer. In the case of the polycrystalline or monocrystalline forms, a and b in Si_aC_b have other values than 0.

Moreover, the growth temperature of the SiC layer according to the
20 present invention is preferably 500°C to $1,100^\circ\text{C}$ since the growth temperature of the $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer is $1,000^\circ\text{C}$ to $1,200^\circ\text{C}$.

Also, the thickness of the SiC layer according to the present invention is preferably 5 Å to 1,000 Å. The larger the thickness, the surface roughness of the $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer to be formed thereon shows a tendency to be severe, however, if the SiC layer becomes
5 excessively thick, the quality of the thin-film will become bad, and as described above, nitrogen will leave from the underlying $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer during the growth of the SiC layer so as to cause the metal aggregation phenomenon. For this reason, the thickness is preferably less than 1,000 Å.

10 The SiC layer according to the present invention may be composed of either a layer n-doped or p-doped with impurities (e.g., n-dopants: periodic table group V elements, and p-dopants: periodic table group III elements) or an n-type layer which has been intentionally undoped but spontaneously formed. The growth of an n-type SiC layer on a p-type $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$
15 ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer corresponds to the use of a tunneling effect.

2) Formation of $\text{Si}_x\text{C}_y\text{N}_z$

To form the SiCN layer in the present invention, DTBSi, a kind of metal
organic source, was used as a silicon source, and CBr_4 , a kind of metal organic
20 source, was used as a carbon source, and ammonia or DMHy was used as a nitrogen source. As described above, the reason why these sources were used is that they are easily thermally decomposed at low temperature so that

they can form the SiCN layer at a lower temperature than that in the use of the existing hydride source, such as CH_4 or SiH_4 . However, in the present invention, sources used in forming the SiCN layer are not limited to DTBSi, CBr_4 , and NH_3 or DMHy. The formation of the SiCN layer on the

5 $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer has an advantage in that the metal aggregation phenomenon on the $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer can be inhibited during the growth of the thin film as compared to the formation of the SiC layer, since ammonia, a hydrazine-based source, or a mixture of ammonia and the hydrazine-based source is used as a

10 nitrogen source. This can prevent nitrogen from leaving from the $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer, since nitrogen groups are continuously supplied during the growth of the SiCN thin film. A disadvantage with the use of the SiCN layer compared to the SiC layer is that the composition thereof is complicated so that the reproduction of the thin film

15 can be slightly reduced as compared to the SiC layer. However, with respect to the $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer to be grown thereon, there is no great difference between the SiC layer and the SiCN layer.

The SiCN layer according to the present invention may be formed on the $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer in various forms,

20 such as amorphous, polycrystalline or monocrystalline forms, depending on conditions. In the case of the polycrystalline or monocrystalline form, c, d and e in $\text{Si}_c\text{C}_d\text{N}_e$ have other values than 0. Also, the growth temperature of the

SiCN layer according to the present invention is preferably 500°C to 1,000°C since the growth temperature of the $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer is generally 1,000°C to 1,200 Å.

Furthermore, the thickness of the SiCN layer according to the present invention is preferably 5 Å to 1,000 Å. The larger the thickness, the surface roughness of the $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer to be grown thereon shows a tendency to be severe, however, if the SiCN layer becomes excessively thick, the quality of the thin film will become bad. For this reason, the thickness is preferably less than 1,000 Å.

The SiCN layer according to the present invention may be composed of either a layer n-doped or p-doped with impurities (e.g., n-dopants: periodic table group V elements, and p-dopants: periodic table group III elements) or an n-type layer which has been intentionally undoped but spontaneously formed. The growth of an n-type SiCN layer on a p-type $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer corresponds to the use of a tunneling effect.

3) Formation of C_fN_g layer

To form the CN layer in the present invention, CBr_4 , a kind of metal organic source, was used as a carbon source, and ammonia, a hydrazine-based source or a mixture of ammonia and the hydrazine-based source was used as a nitrogen source. The use of CBr_4 provides an advantage in that it is easily thermally decomposed at low temperature so that

it can form a CN layer at low temperature. However, in the present invention, a carbon source used in forming the CN layer is not limited only to CBr_4 . The CN layer has an advantage in that a problem, such as the leaving of nitrogen from the underlying $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer, does not occur during the growth of the CN layer unlike the SiC layer, since the nitrogen sources (ammonia and hydrazine-based source) is continuously supplied during the growth of the thin film.

The CN layer according to the present invention may be formed on the $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer in various forms, such as amorphous, polycrystalline or monocrystalline forms, depending on conditions. These crystalline forms may vary depending on conditions, such as the growth temperature of the CN layer. In the case of the polycrystalline or monocrystalline form, f and g in the C_fN_g layer have other values than 0.

Also, the growth temperature of the C_fN_g layer according to the present invention is preferably 500°C to $1,000^\circ\text{C}$ since the growth temperature of the $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer is generally $1,000^\circ\text{C}$ to $1,200^\circ\text{C}$.

Furthermore, the thickness of the C_fN_g layer according to the present invention is preferably 5\AA to $1,000\text{\AA}$. The larger the thickness, the roughness of the $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer to be grown thereon shows a tendency to be severe, however, if the SiCN layer becomes excessively thick, the quality of the thin-film will become bad. This is

mainly because of the lattice mismatch between the C_rN_g layer and the $Al(x)Ga(y)In(1-x-y)N$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer.

The C_rN_g layer according to the present invention may be composed of either a layer n-doped or p-doped with impurities (e.g., n-dopants: periodic
5 table group V elements, and p-dopants: periodic table group III elements) or an n-type layer which has been intentionally undoped but spontaneously formed. The growth of an n-type C_rN_g layer on a p-type $Al(x)Ga(y)In(1-x-y)N$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer corresponds to the use of a tunneling effect.

10 The first layer composed of the n-type or p-type silicon carbide layer (SiC), silicon carbon nitride (SiCN) or carbon nitride (CN) layer as described above can be formed as a uniform or nonuniform layer as shown in FIGs. 3 to 5.

The formation of this uniform or nonuniform layer can be determined depending on the thickness and thin-film quality of the first layer. For example,
15 the first layer is made of materials completely different from the underlying $AlInGaN$ layer, and thus, if it is grown in a single crystalline form, it can be grown into a nonuniform layer. FIG. 3 shows the protrusions 32 of the $AlInGaN$ layer formed with roughness between the films of the first layer 31 discontinuously formed. FIG. 4 shows the protrusions 34 of the $AlInGaN$ layer
20 formed with roughness on the first layer 33 uniformly formed. FIG. 5 shows the protrusions 32 and 34 of the $AlInGaN$ layer formed with roughness between and on the films of the first layer 35. The inventive structures with roughness

may include all the three cases of FIG. 3 to FIG. 5.

Characteristics in forming the first layer composed of the n-type or p-type silicon carbide (SiC), silicon carbon nitride (SiCN) or carbon nitride (CN) layer on the P-AlInGaN layer of the existing III-nitride and forming the second
5 layer composed of the $P-Al_xGa_yIn_zN$ layer on the first layer will now be described.

As shown in FIGs. 3 to 5, when the second layer 21 made of $Al(x)Ga(y)In(1-x-y)N$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) is grown on the first layer composed of the n-type or p-type silicon carbide (SiC), silicon carbon nitride
10 (SiCN) or carbon nitride (CN) layer, the second layer 21 will aggregate in an island configuration due to a difference in material characteristics (e.g., lattice constant) between the two layers. This is easily understood when imagining a phenomenon where if water is spilt on a material which is not easily wetted, water will aggregate in drops.

15 When the first layer is the nonuniform layer 31 as shown in FIG. 3, the second layer composed of the $Al(x)Ga(y)In(1-x-y)N$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer will be grown with respect to grooves while forming roughness. When the first layer is the uniform layer 33, initial seeds of $Al(x)Ga(y)In(1-x-y)N$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) will be randomly formed
20 on the first layer, and roughness will be formed in the configuration of the islands 34 with respect to the seeds.

The surface roughness can be adjusted according to the thin-film

thickness of the second layer, and the thickness of the first layer also performs an important role in the surface roughness.

FIG. 6 shows scanning electron microscope (SEM) images of the $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer with roughness (protrusions and/or depressions) formed according to the present invention. The light portions in FIG. 6 is the $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer formed in an island configuration, and the dark portions are portions corresponding to the first layer. These portions have a size ranging from several thousands Angstroms to 10 microns and various shapes. The size and shape of these portions may vary depending on the thickness of the grown $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$) layer. As described above, the roughness so formed can maximize the external quantum efficiency of light emitted from the active layer.

FIGs. 7 and 8 show surface images measured by AFM equipment for roughness formed according to the present invention. The AFM equipment allows the three-dimensional image of a narrower region to be precisely obtained unlike the above-mentioned SEM equipment. FIG. 7 shows a surface image of a $1.5 \times 1.5 \text{ }\mu\text{m}$ region with roughness. As can be seen in FIG. 7, the roughness portions have a size of several hundreds Angstroms to 10 microns and various shapes. FIG. 8 shows a cross-sectional profile of the image shown in FIG. 7. As can be seen in FIG. 8, the cross-section of the fine roughness structures is trapezoid in shape. Also, the inclined angle of the

roughness surface is 10° to 60° or greater. Thus, this inclined surface will change the directional angle of incident light, thus increasing the external quantum efficiency of the light.

FIG. 9 is a graph showing an increase in light output with an increase in the thickness of the first layer 20 of the inventive light emitting device with roughness. As can be seen in FIG. 9, the maximum light output of the inventive light emitting device is at least two times higher than the existing light emitting device with no roughness.

FIG. 10 shows scanning electron microscope photographs showing the surface roughness as a function of the thickness of the first layer. The first photograph is a surface photograph of a normal light emitting diode, the second photograph is for the case where the SiC layer has been formed in a thickness of about 10 Angstroms, and the third photograph is for the case where the SiC layer has been formed in a thickness of about 20 Angstroms. In this regard, the thickness of the SiC layer is expected based on the growth rate, and can slightly differ from an actual value. As shown in FIG. 10, the larger the thickness of the SiC layer, the surface roughness increases.

The light emitting diode with roughness according to the present invention can be fabricated by methods described in the following examples.

Example 1

A III-nitride compound semiconductor light emitting device is fabricated by forming a first layer composed of an n-type silicon carbide (Si_aC_b), n-type silicon carbon nitride ($\text{Si}_c\text{C}_d\text{N}_e$) or n-type carbon nitride (C_fN_g) (a, b, c, d,

e, f, and g are not 0) layer on a p-type GaN layer forming the existing upper p-side electrode contact layer in a thickness of 5 Å to 1000 Å and growing a second layer composed of a p-type GaN layer on the first layer in a thickness of 100 Å to 5,000 Å.

5 Example 2

A III-nitride compound semiconductor light emitting device is fabricated by forming a first layer composed of a p-type silicon carbide (Si_aC_b), p-type silicon carbon nitride ($\text{Si}_c\text{C}_d\text{N}_e$) or p-type carbon nitride (C_fN_g) (a, b, c, d, e, f, and g are not 0) layer on a p-type GaN layer forming the existing upper
10 p-side electrode contact layer in a thickness of 5 Å to 1000 Å and growing a second layer composed of a p-type GaN layer on the first layer in a thickness of 100 Å to 5,000 Å.

 Example 3

A III-nitride compound semiconductor light emitting device is
15 fabricated by forming a first layer composed of an n-type silicon carbide (Si_aC_b), n-type silicon carbon nitride ($\text{Si}_c\text{C}_d\text{N}_e$) or n-type carbon nitride (C_fN_g) (a, b, c, d, e, f, and g are not 0) layer on a p-type GaN layer forming the existing upper p-side electrode contact layer in a thickness of 5 Å to 1000 Å and growing a second layer composed of a p-type GaN layer on the first layer in a thickness
20 of 100 Å to 5,000 Å and growing a third layer composed of an n-type GaN layer on the second layer in a thickness of 5 Å to 200 Å. In this case, the third layer is a layer formed using the concept of a tunneling effect.

Example 4

A III-nitride compound semiconductor light emitting device is fabricated by forming a first layer composed of a p-type silicon carbide (Si_aC_b), p-type silicon carbon nitride ($\text{Si}_c\text{C}_d\text{N}_e$) or p-type carbon nitride (C_fN_g) (a, b, c, d, e, f, and g are not 0) layer on a p-type GaN layer forming the existing upper p-side electrode contact layer in a thickness of 5 Å to 1000 Å growing a second layer composed of a p-type GaN layer on the first layer in a thickness of 100 Å to 5,000 Å and growing a third layer composed of an n-type GaN layer on the second layer in a thickness of 5 Å to 200 Å. In this case, the third layer is a layer formed using the concept of a tunneling effect.

Example 5

Each of the light emitting devices described in Examples 1, 2, 3 and 4 is characterized in that the surface layer electrode is made of any one selected from the group consisting of nickel, gold, silver, chrome, titanium, platinum, palladium, rhodium, iridium, aluminum, tin, ITO, indium, tantalum, copper, cobalt, iron, ruthenium, zirconium, tungsten, and molybdenum.

Also, the second and third layers may be made of $\text{Al}(x)\text{Ga}(y)\text{In}(1-x-y)\text{N}$ ($0 \leq x \leq 1$, $0 \leq y \leq 1$, $0 \leq x+y \leq 1$), in addition to GaN.

One that a person skilled in the art should take notice in understanding the means of the thickness of the first and second layers described in Examples is how the thickness of the second layer is to be understood when the first layer is a nonuniform layer, in which case the thickness of the second

layer may be understood to be a thickness from the surface of the first layer, assuming that the first layer is a uniform layer. The addition of this description by the applicant is to avoid useless disputes in analysis related with the thickness of the first and second layers, since growing the first layer in the form
5 of an ideal uniform layer is not easy in view of the characteristics of the present invention using a difference in material characteristics between the carbon-containing compound and the III-nitride compound semiconductor and also not essential in the present invention.

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